

# Freestanding Nanoengineered [001] Preferentially Oriented TiO<sub>2</sub> Nanosheets—Graphene Planarly Aligned Nanohybrids with Enhanced Li-Storage Properties

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A freestanding film electrode comprising face-to-face alternately stacked 2D TiO<sub>2</sub> nanosheets and reduced graphene oxide (rGO) is designed and fabricated as a binder-free anode for lithium-ion batteries. The ultrathin TiO<sub>2</sub> nanosheets, with (001) facets exposed were face-to-face hetero-assembled onto the conductive rGO, allowing Li<sup>+</sup> to swiftly diffuse into the TiO<sub>2</sub> nanosheets across the (001) facets with the lowest barrier and a short transfer distance. The configuration of this planar-aligned

electrode leads to superior lithium performance, which delivers a large capacity of 132.3 mA h g<sup>-1</sup> with a capacity retention of 89% over 1000 cycles at a high rate of 20 C (1 C = 168 mA g<sup>-1</sup>). This work proves that controlling the arrangement orientation of electrode materials with an anisotropy Li<sup>+</sup> diffusion barrier could remarkably improve the electrochemical properties, for promising applications in high-power energy-storage devices.

## 1. Introduction

The increasing market of portable devices, electrical vehicles, and grid energy storage has spurred rapid development of rechargeable lithium ion batteries (LIBs).<sup>[1–3]</sup> Anatase titanium dioxide (TiO<sub>2</sub>) that is an effective photocatalyst in photocatalytic decomposition of organic pollutants and water splitting is also a promising anode material in LIBs due to the high operating potential relative to graphite, small volume change (3–4%) upon lithiation, low toxicity, environmental benignity, and widespread availability.<sup>[4–7]</sup> However, the low ionic diffusivity and poor electrical conductivity of TiO<sub>2</sub> result in low reversible capacity and limited rate performance pertaining to Li storage.<sup>[7–10]</sup> Hence, attempts have been made to overcome these hurdles and the main strategies include combining TiO<sub>2</sub> with conductive carbonaceous materials and fabricating nanostructured TiO<sub>2</sub>.<sup>[10–15]</sup> For example, graphene, a typical two-dimensional (2D) carbonaceous material, is a promising host to load active materials in LIBs on account of the excellent electrical conductivity, mechanical strength, and large specific surface.<sup>[15–20]</sup> Various TiO<sub>2</sub>/graphene composites such as nano-

particles/N-doped graphene foam,<sup>[15]</sup> electrospinning TiO<sub>2</sub>/graphene composites<sup>[20]</sup> and mesoporous TiO<sub>2</sub> nanocrystals/graphene aerogels<sup>[21]</sup> possess enhanced electrochemical properties compared to pure TiO<sub>2</sub> due to the improved electrical conductivity of the composites.

Tuning the morphology of TiO<sub>2</sub> on the nanoscale is another strategy to improve the Li storage properties of TiO<sub>2</sub> by shortening the Li<sup>+</sup> diffusion distance or reducing the Li<sup>+</sup> diffusion barrier. It has been reported that Li<sup>+</sup> diffusion and electron migration across the (001) facet of anatase TiO<sub>2</sub> has a lower energy barrier than other crystal facets such as (101).<sup>[22–27]</sup> Thus, 2D anatase TiO<sub>2</sub> nanosheets with exposed (001) facet should exhibit improved reaction kinetics and rate performance in Li storage due to the larger reactive surface and lower Li<sup>+</sup> diffusion barrier across the (001) facet. Generally, TiO<sub>2</sub>-based anodes are prepared by mixing 2D TiO<sub>2</sub> nanosheets with graphene or conductive agents as well as nonconductive organic binders followed by pressing them on the Cu foil. Thus, the 2D TiO<sub>2</sub> nanosheets with the exposed (001) facet are not fully parallel to the current collector in these anodes.<sup>[22,25–27]</sup> Moreover, the mass loading of TiO<sub>2</sub> in those anodes is generally low due to the existing of inactive materials such as binders and copper current collector. Hence, the electrochemical performance of the 2D TiO<sub>2</sub> is not fully utilized in these anodes. If these TiO<sub>2</sub> nanosheets are face-to-face hetero-assembled onto the graphene current collector to form a planarly aligned TiO<sub>2</sub>/graphene freestanding film acting as anodes, the excellent Li storage properties such as high rate capability and large reversible capacity will be expected. For the freestanding planarly aligned TiO<sub>2</sub>-reduced graphene oxide (rGO) hybrid electrode, since the diffusion direction of the Li<sup>+</sup> is in accordance with the direction of the electrostatic field that is perpendicular to the surface of the rGO current collector, Li<sup>+</sup> can swiftly diffuse into the TiO<sub>2</sub> nanosheets across the (001) facets with a low barrier and short diffusion length and the

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electrons can transfer swiftly along the highly conductive rGO nanosheets. Such configuration of planarly aligned TiO<sub>2</sub>-rGO paper electrode takes the advantage of 2D TiO<sub>2</sub> and graphene nanosheets, resulting in enhanced electrochemical properties. However, this superior hybrid film electrode has not been realized.

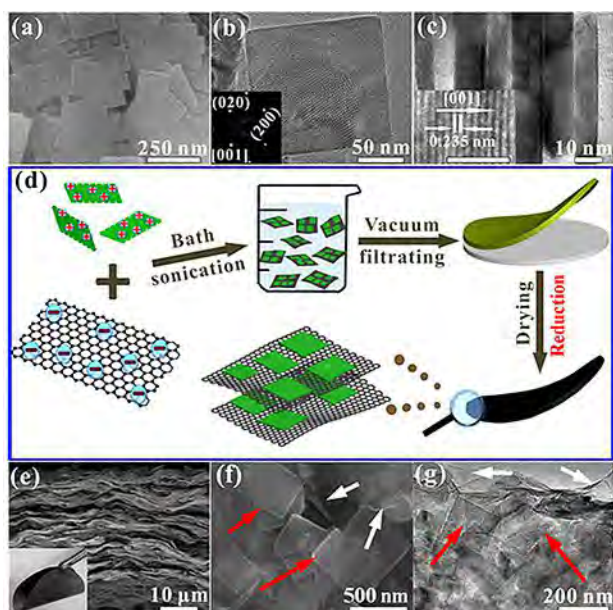
In this work, a freestanding film electrode comprising face-to-face alternately stacked 2D TiO<sub>2</sub> nanosheets and rGO is fabricated as a binder free, large-capacity, high-rate, and long cycling-life anode for LIBs. The spectacular configuration of this planarly-aligned electrode led to superior performances than other composite electrodes previously reported. The scalable fabrication process is simple and can be extended to other anode and cathode materials for LIBs.

## 2. Results and Discussion

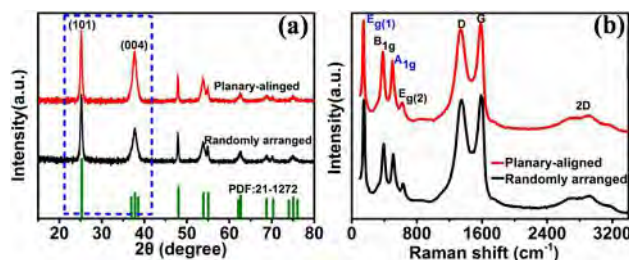
The TiO<sub>2</sub> nanosheets with exposed (001) facets are prepared by a modified hydrothermal technique using Ti(OC<sub>4</sub>H<sub>9</sub>)<sub>4</sub> as the precursor.<sup>[28,29]</sup> Figure 1a depicts the field-emission scanning electron microscopy (FE-SEM) image of the hydrothermal product synthesized at 200 °C, revealing well-defined sheet-shape structures with rectangular outlines. Figures 1b and 1c present the top-view and cross-sectional transmission electron microscopy (TEM) image of the TiO<sub>2</sub> nanosheets which have a rectangular outline and the side length and thickness are about 200 and 10 nm, respectively. The selected-area electron diffrac-

tion (SAED) pattern (inset in Figure 1b) and high-magnification TEM (HR-TEM) image (Figure 1c) indicate that the top and bottom facets of the nanosheets are the (001) planes and the lattice spacing parallel to the top and bottom facets is about 0.235 nm, corresponding to the (001) planes of anatase TiO<sub>2</sub>.<sup>[29]</sup> The X-ray diffraction (XRD) pattern (JCPDS No. 21-1272) and Raman scattering spectrum<sup>[15]</sup> of the nanosheets are depicted in Figure S1, which can be ascribed to anatase TiO<sub>2</sub>. The percentage of the exposed (001) facet ratio in the anatase TiO<sub>2</sub> nanosheets is as large as 91 % (Figure S2).

The preparation procedures for the freestanding planarly aligned TiO<sub>2</sub>-rGO paper electrode (denoted as planarly aligned) are schematically illustrated in Figure 1d. The TiO<sub>2</sub> and rGO nanosheets were prone to layer-by-layer self-assembly onto the filter membrane to afford the alternately stacked 2D TiO<sub>2</sub> nanosheets and rGO hybrid film<sup>[30–32]</sup> (See details in experimental section), the thickness of GO is 2.5–3 nm with the number of 8–10 layers (Figure S3). The cross-sectional SEM image (Figure 1e) discloses that the freestanding planarly aligned TiO<sub>2</sub>-rGO paper electrode has a thickness of about 30 μm. Thermal analysis (Figure S4) reveals that the content of TiO<sub>2</sub> nanosheets and rGO in this film is about 93.1 wt.% and 6.9 wt.%, respectively. The digital image in the inset of Figure 1e demonstrates good flexibility and mechanical integrity of this film although it does not contain any binders. The FE-SEM image (Figure 1f) and TEM image (Figure 1g) clearly demonstrate that the thin TiO<sub>2</sub> nanosheets are face-to-face hetero-assembled onto the rGO nanosheets forming alternating stacked TiO<sub>2</sub> and rGO hybrid structure. The energy dispersive spectrometer (EDS) maps show uniform distribution of Ti, O, and C, confirming the homogeneous TiO<sub>2</sub>-rGO hybrid structure (Figure S5). For comparison, TiO<sub>2</sub> nanosheets are also physically mixed with rGO to produce randomly arranged TiO<sub>2</sub>-rGO paper electrode (denoted as randomly arranged electrode), in which some of TiO<sub>2</sub> nanosheets are not parallel to the rGO current collector but upright on the rGO as shown in Figure S6. Figures 2a–b are the XRD pattern and Raman spectra of planarly aligned and randomly arranged TiO<sub>2</sub>-rGO film electrode. Although the XRD spectra demonstrate the same peak positions of anatase TiO<sub>2</sub> (JCPDS No. 21-1272), the intensity ratio of I<sub>(004)</sub>/I<sub>(101)</sub> for planarly aligned TiO<sub>2</sub>-rGO (0.69) is higher than that of randomly arranged TiO<sub>2</sub>-rGO paper electrode (0.51). Since the upper and top and bottom faces of the nanosheets are the (001) planes and the side faces consist of



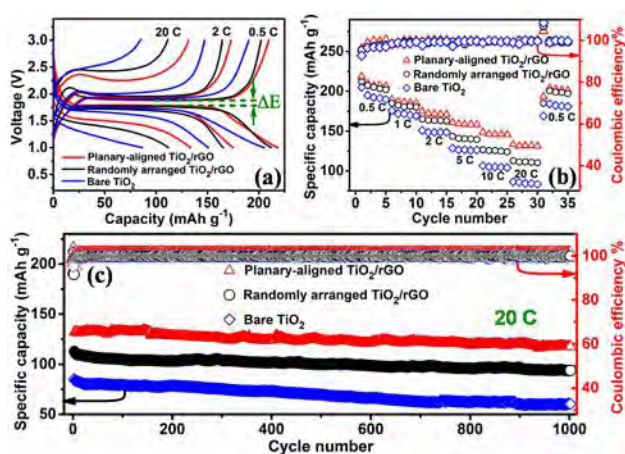
**Figure 1.** a) SEM image of TiO<sub>2</sub> nanosheets. b) TEM image of a typical TiO<sub>2</sub> nanosheet. c) cross-sectional TEM view of a bundle of TiO<sub>2</sub> nanosheets, the inset is the high-resolution TEM image of one nanosheet. d) Schematic illustration of the preparation procedures of the planarly aligned TiO<sub>2</sub>-rGO paper electrode. The side-view SEM image (e), top-view SEM image (f), and TEM image (g) of planarly aligned TiO<sub>2</sub>-rGO paper electrode. The red arrows indicate the TiO<sub>2</sub> nanosheets and the white arrows indicate the rGO. The inset digital image in (e) indicates that the paper electrode has robust flexibility.



**Figure 2.** XRD patterns (a) and Raman spectra (b) of planarly aligned and randomly arranged TiO<sub>2</sub>-rGO paper electrode.

(101) planes, the higher  $I_{(004)}/I_{(101)}$  value suggests that more 2D  $\text{TiO}_2$  are face-to-face assembled on the rGO forming planarly aligned  $\text{TiO}_2$ -rGO paper electrode.<sup>[33]</sup> The Raman peaks show the typical Raman shifts of anatase  $\text{TiO}_2$ <sup>[15,34]</sup> and rGO. The  $E_{g(1)}$  ( $144\text{ cm}^{-1}$ ) peak is ascribed to the symmetric stretching vibration of O–Ti–O of  $\text{TiO}_2$  and the  $A_{1g}$  ( $514\text{ cm}^{-1}$ ) peak is attributed to the asymmetric bending vibration of O–Ti–O. The bonding modes on the (101) surface are the saturated Ti and O mode contributing to the symmetric stretching vibration<sup>[28]</sup> and the unsaturated Ti and O mode corresponding to asymmetric bending vibration,<sup>[34]</sup> however (001) plane comprises of the unsaturated Ti and O bonding modes. The higher intensity ratio of  $I_{A_{1g}}/I_{E_{g(1)}}$  for planarly aligned  $\text{TiO}_2$ -rGO (0.53) compared with randomly arranged  $\text{TiO}_2$ -rGO (0.39) means the more 2D  $\text{TiO}_2$  are face-to-face assembled on the rGO for planarly aligned  $\text{TiO}_2$ -rGO paper electrode, which is agreement with the XRD results.

The electrochemical properties of the  $\text{TiO}_2$  nano-sheets based electrode are evaluated using coin-like 2025 batteries with metallic Li as the counter electrode in a voltage range of 1.0–3.0 V vs.  $\text{Li}^+/\text{Li}$ . Figure 3a displays the galvanostatic charge–



**Figure 3.** a) The galvanostatic charge–discharging profiles at 0.5, 2, and 20 C ( $1\text{ C} = 168\text{ mA g}^{-1}$ ). b) The rate performance with coulombic efficiency. c) cycling performance and coulombic efficiency at a rate of 20 C of  $\text{TiO}_2$  nanosheets-based electrodes.

ing-discharging (GCD) profiles of the planarly aligned, randomly arranged film and bare  $\text{TiO}_2$  nano-sheets electrodes at current rates of 0.5, 2 and 20 C ( $1\text{ C} = 168\text{ mA g}^{-1}$ ). At a low rate of 0.5 C, the distinct plateaus at 1.77 and 1.90 V correspond to typical discharging ( $\text{Li}^+$  insertion) and charging ( $\text{Li}^+$  extraction) processes in the anatase  $\text{TiO}_2$ .<sup>[13–15]</sup> At this rate, these three electrodes deliver specific capacities of 218.5, 211.9 and 205.0  $\text{mA h g}^{-1}$  with first coulombic efficiency of 96.0%, 95.4%, and 92.8%, respectively. The initial coulombic efficiency of our  $\text{TiO}_2$  and  $\text{TiO}_2$ -rGO paper electrode is much higher than most other transition metal oxides, alloys and carbon based anodes.<sup>[35–37]</sup> Generally, the large initial capacity loss is a common phenomenon for metal oxide-based anodes due to the formation of unstable solid-electrolyte interphase (SEI) and poorly conductive  $\text{Li}_2\text{O}$ . The large initial coulombic efficiency of

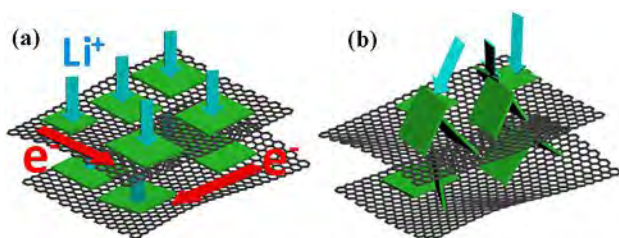
planarly aligned  $\text{TiO}_2$ -rGO electrode is mainly due to the two following reasons: (a) The  $\text{TiO}_2$  has a high lithium ion insertion/extraction potential at about 1.70 V, which avoids the formation of unstable SEI; (b)  $\text{TiO}_2$  stores Li via the insertion reaction mechanism,  $\text{TiO}_2 + x\text{Li}^+ + xe^- \rightleftharpoons \text{Li}_x\text{TiO}_2$ , while most other metal oxide anodes store Li via conversion reaction mechanism,  $M_x\text{O}_y + 2y\text{Li}^+ + 2ye^- \rightleftharpoons xM + y\text{Li}_2\text{O}$  ( $M\text{O}_x$ , where  $M = \text{Fe}, \text{Co}, \text{Ni}, \text{etc.}$ ), the formation of low conductive of  $\text{Li}_2\text{O}$ , causing large initial capacity loss. The large capacity of planarly aligned  $\text{TiO}_2$ -rGO film should mainly stem from  $\text{TiO}_2$  and the capacity contribution from rGO is negligible, as depicted in Figure S7. It's worth noting that the capacity value at 0.5 C exceeds the theoretical capacity of  $\text{TiO}_2$  ( $168\text{ mA h g}^{-1}$ ) due to the large surface capacitance of nano-sized  $\text{TiO}_2$ .<sup>[15,19]</sup> The planarly aligned  $\text{TiO}_2$ -rGO hybrid film exhibits high reversibility, when the current density is further increased to 2 and 20 C, the charging and discharging plateaus of planarly aligned  $\text{TiO}_2$ -rGO hybrid film are clearly observed.  $\Delta E$  is the difference between the charging and discharging potential plateaus reflecting the degree of polarization at different rates.<sup>[38]</sup> The smaller  $\Delta E$  is, the better is the rate capability of the battery. As shown in Figure 3a and Figure S8, the planarly aligned  $\text{TiO}_2$ -rGO paper electrode has the lowest  $\Delta E$ , especially at the high rate of 20 C. The  $\Delta E$  of the planarly aligned  $\text{TiO}_2$ -rGO paper electrode is only 0.76 V, which is smaller than that of the randomly arranged  $\text{TiO}_2$ -rGO film electrode (0.94 V) and the bare  $\text{TiO}_2$  electrode (1.21 V), implying faster  $\text{Li}^+$  diffusion and electron transfer at the large current density. The rate capabilities of these three electrodes are presented in Figure 3b. The planarly aligned  $\text{TiO}_2$ -rGO paper showed the best rate performance, especially at high rate. At high current rates of 5, 10 and 20 C, the planarly aligned  $\text{TiO}_2$ -rGO paper electrode delivers capacities of 160.7, 148.6, and 132.3  $\text{mA h g}^{-1}$ , respectively, which are much larger than those of the randomly arranged  $\text{TiO}_2$ -rGO paper (143.3, 126.5, 112.7  $\text{mA h g}^{-1}$ , respectively) and bared  $\text{TiO}_2$  (128.3, 106.4, 86.3  $\text{mA h g}^{-1}$ , respectively). The long-term cycling stability is presented in Figure 3c. All electrodes show good cycling stability at a high rate of 20 C with high coulombic efficiency closed to 100%. However, the planarly aligned  $\text{TiO}_2$ -rGO paper electrode shows a largest capacity of 132.3  $\text{mA h g}^{-1}$  with capacity retention of 89% over 1,000 cycles at a rate of 20 C, whereas the randomly arranged  $\text{TiO}_2$ -rGO paper delivers a capacity of 83.3  $\text{mA h g}^{-1}$  with capacity retention of 83.3% and the bare  $\text{TiO}_2$  delivers a capacity of 85.2  $\text{mA h g}^{-1}$  with capacity retention of 71.0%. After 1000 cycles, the charging and discharging plateau remain intact, further indicating high reversibility (Figure S9) of the planarly aligned  $\text{TiO}_2$ -rGO paper electrode. At a low rate of 1 C, the planarly aligned  $\text{TiO}_2$ -rGO paper also shows best cycling performance than the other two electrodes (Figure S10). The superior cyclability of the planarly aligned  $\text{TiO}_2$ -rGO paper electrode should be attributed to the high electrochemical stability of  $\text{TiO}_2$  nanosheets and hybrid electrode design by taking the advantage of 2D  $\text{TiO}_2$  and graphene nanosheets. In the planarly aligned  $\text{TiO}_2$ -rGO free-standing hybrid film, no polymer binders and conductive additives are utilized, consequently reducing the undesirable interface, eliminating the associated overpotential and improv-

ing the utilization of active materials. Therefore, large capacity, super rate capability and prolonged cycle life are achieved for the planarly aligned TiO<sub>2</sub>-rGO hybrid film electrode, which are superior to those previously reported TiO<sub>2</sub>-based electrodes<sup>[14–16,19,39–48]</sup> as showed in Table 1.

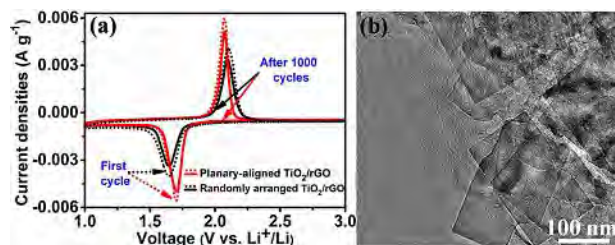
**Table 1.** The electrochemical performances of Planarly aligned TiO<sub>2</sub>-rGO paper electrode and other TiO<sub>2</sub>-based electrodes.

Materials	Current density [C]	Performance [mAh g <sup>-1</sup> ]	Ref.
TiO <sub>2</sub> /Ag hollow fibers	1	110 @ 50 cycles	[14]
Graphene/TiO <sub>2</sub>	20	96 @ 80 cycles	[15]
TiO <sub>2</sub> -Graphene	20	110 @ 30 cycles	[16]
TiO <sub>2</sub> -rGO	20	108 @ 150 cycles	[19]
Self-Assembled TiO <sub>2</sub>	4	110 @ 40 cycles	[22]
TiO <sub>2</sub> Hollow Spheres	20	81.4 @ 100 cycles	[39]
Carbon /TiO <sub>2</sub>	15	29.3 @ 30 cycles	[40]
TiO <sub>2</sub> spheres	10	110 @ 60 cycles	[41]
Nanoporous-TiO <sub>2</sub>	20	94 @ 70 cycles	[42]
TiO <sub>2</sub> Cubes	2	96 @ 30 cycles	[43]
Carbon Foams-TiO <sub>2</sub>	12	113 @ 20 cycles	[44]
TiO <sub>2</sub> Mesocrystals	1	151.9 @ 60 cycles	[45]
Hierarchical TiO <sub>2</sub>	20	99 @ 30 cycles	[46]
TiO <sub>2</sub> Nanosheets	10	101.9 @ 100 cycles	[47]
3D graphene/TiO <sub>2</sub>	12	100 @ 40 cycles	[48]
<b>Planarly aligned TiO<sub>2</sub>-rGO paper</b>	<b>1</b>	<b>168 @ 400 cycles</b>	<b>this work</b>
	<b>20</b>	<b>110 @ 1000 cycles</b>	

The above results clearly indicate that the rational hybrid electrode design of TiO<sub>2</sub> nanosheets and rGO plays an important role in the electrochemical properties. In the planarly aligned TiO<sub>2</sub>-rGO paper electrode, rGO is used as a current collector and the (001) facets of the TiO<sub>2</sub> nanosheets are face-to-face assembled on the rGO and parallel to the surface of the rGO thus forming an anisotropic electrode, as schematically shown in Figure 1d and Figure 4a. Thus, Li<sup>+</sup> diffusing along the



**Figure 4.** Schematic diagrams of lithium ion diffusion for the planarly aligned TiO<sub>2</sub>-rGO paper electrode (a) and randomly arranged TiO<sub>2</sub>-rGO paper electrode (b). For the planarly aligned TiO<sub>2</sub>-rGO paper electrode in (a), Li<sup>+</sup> can fully intercalate into the anatase TiO<sub>2</sub> across the (001) facet with a low barrier and short transfer distance along the electric field.



**Figure 5.** a) Cyclic voltammetry curves of the planarly aligned TiO<sub>2</sub>-rGO paper electrode and randomly arranged paper electrode at the scanning rate of 0.1 mV s<sup>-1</sup> before (dot-line) and after (solid-line) 1000 cycles at the rate of 20 C. b) TEM image of the planarly aligned TiO<sub>2</sub>-rGO paper electrode after 1000 cycles.

electric field can fully intercalate into the anatase TiO<sub>2</sub> across the (001) facet with a low barrier and electrons can move quickly to the TiO<sub>2</sub> nanosheets from the rGO thereby yielding excellent rate capability. However, if the TiO<sub>2</sub> nanosheets and rGO are physically mixed, some TiO<sub>2</sub> nanosheets are not parallel to the rGO nanosheets and even standing upright on the rGO nanosheets as schematically shown Figure 4b and Figure S6. In this case, along the direction of the electric field in LIBs, Li<sup>+</sup> also insert into the (101) facet during charging–discharging, thus presenting a higher barrier for Li<sup>+</sup> diffusion and longer diffused distance than the (001) facets.

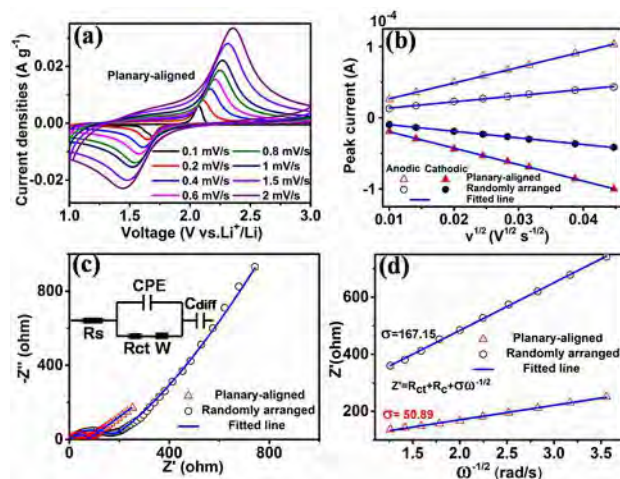
To further understand the improved high-rate performance and long cycle performance of the planarly aligned TiO<sub>2</sub>-rGO paper electrode. The cyclic voltammetry profiles at a fixed scan rate of 0.1 mV s<sup>-1</sup> before and after 1000 cycles at the rate of 20 C were performed. CV curves in Figure 5a exhibit two sharp peaks at 1.7 V and 2.1 V corresponding to the Li<sup>+</sup> insertion/extraction reaction in anatase TiO<sub>2</sub>, agreeing well with GCD curves in Figure 3a. The potential interval between the anodic and cathodic peaks of the planarly aligned TiO<sub>2</sub>-rGO paper electrode is about 360 mV, which is smaller than that of the randomly arranged TiO<sub>2</sub>-rGO film (460 mV) indicative of better rate capability. Moreover, the peak intensity of the former is larger than that of the latter, demonstrating faster Li<sup>+</sup> reaction kinetics and larger capacity. After 1000 cycles, the peak shapes and positions from CV curves remain the same (Figure 5), indicating high stability of the paper electrode during long cycle. It also can find that after long cycle at high rate, the redox potential interval acquired from planarly aligned TiO<sub>2</sub>-rGO paper (365 mV) is still smaller than that of the randomly arranged TiO<sub>2</sub>-rGO film (455 mV) indicative of better rate capability of planarly aligned TiO<sub>2</sub>-rGO paper. The morphology of the TiO<sub>2</sub> nanosheets has no change after 1000 cycles (Figure 5b), also revealing high structural stability of TiO<sub>2</sub> nanosheets.

Figure 6a and Figure S11 showed the CV curves of two different electrodes at different scanning rates. Two peaks corresponding to the Li<sup>+</sup> insertion/extraction reaction in anatase TiO<sub>2</sub> were found, agreeing well with GCD curves in Figure 3a. When the scanning rate is further increased to 1 and 2 mV s<sup>-1</sup>, the charging and discharging peaks of planarly aligned TiO<sub>2</sub>-rGO hybrid film are clearly observed, suggesting excellent chemical reversibility. ΔE can also be calculated based

on the difference between the charging and discharging peaks also reflect the degree of polarization at different scan rates. The smaller  $\Delta E$  is, the better rate capability of the battery is. As shown in Figure 6a and Figure S11, the  $\Delta E$  became larger with the increasing scanning rate, but the accelerated speed of planarly aligned  $\text{TiO}_2$ -rGO hybrid is smaller, and the  $\Delta E$  of this  $\text{TiO}_2$ -rGO electrode (0.9 V) at the high scan rate of  $2 \text{ mVs}^{-1}$  is smaller than that of the randomly arranged  $\text{TiO}_2$ -rGO film electrode (1.06 V), implying faster  $\text{Li}^+$  diffusion and electron transfer at a large scanning rate although they have the same composition. This result is consistent with the results of GCD profiles (Figure 3a). Figure 6b illustrates the relationship between the peak current ( $I$ ) and square root of different scanning rates in the range of  $0.1$ – $2 \text{ mVs}^{-1}$  (details shown in Figure S11). The good linear relationship indicates that the Li insertion/extraction reaction in anatase  $\text{TiO}_2$  in both samples is diffusion-controlled.<sup>[49]</sup> Hence, the  $\text{Li}^+$  diffusion rate has an important effect on the Li storage properties of the hybrid electrodes. The  $\text{Li}^+$  diffusion coefficient could be estimated according to the Randles-Sevcik equation [Eq. (1)].<sup>[50]</sup>

$$I = 2.69 \times 10^5 n^{3/2} A D^{1/2} C v^{1/2} \quad (1)$$

where  $n$  is the number of electrons in the charge transfer step (for anatase  $\text{TiO}_2$ ,  $n=0.5$ ),<sup>[51]</sup>  $A$  is the surface area of the electrode ( $\text{cm}^2$ ),  $D$  is the  $\text{Li}^+$  diffusion coefficient in the different films at 298 K ( $\text{cm}^2 \text{ s}^{-1}$ ),  $C$  is the molar concentration of  $\text{Li}^+$  in anatase  $\text{TiO}_2$  ( $2.5 \times 10^{-2} \text{ mol cm}^{-3}$ ),<sup>[51]</sup> and  $v$  is the scanning rate ( $\text{Vs}^{-1}$ ). The  $\text{Li}^+$  diffusion coefficients for insertion and extraction reactions from the planarly aligned  $\text{TiO}_2$ -rGO paper are calculated to be  $8.04 \times 10^{-13} \text{ cm}^2 \text{ s}^{-1}$  and  $7.36 \times 10^{-13} \text{ cm}^2 \text{ s}^{-1}$ , respectively, which are larger than those of the randomly arranged  $\text{TiO}_2$ -rGO paper ( $1.30 \times 10^{-13} \text{ cm}^2 \text{ s}^{-1}$  and  $1.16 \times 10^{-13} \text{ cm}^2 \text{ s}^{-1}$ ), indicating faster  $\text{Li}^+$  diffusion in the planarly aligned  $\text{TiO}_2$ -rGO hybrid paper electrode. Figure 6c shows the EIS profiles after 1000<sup>th</sup> cycle and the insets are the equivalent circuits. The electrochemical impedance spectra (EIS) of planarly aligned  $\text{TiO}_2$ -rGO hybrid before and after 1000<sup>th</sup> cycle are depicted Figure S12. The semicircle across the high-medium frequency region represents the charge transfer impedance ( $R_{ct}$ ) and  $W$  represents the Warburg impedance.<sup>[49,50]</sup> After 1000<sup>th</sup> cycle, the planarly aligned  $\text{TiO}_2$ -rGO hybrid paper electrode shows a smaller  $R_{ct}$  of  $69.3 \Omega$  compared to the physically mixed paper electrode ( $166.6 \Omega$ ), implying higher conductivity and faster electron transport during  $\text{Li}^+$  insertion/extraction. Moreover, the  $R_{ct}$  of the planarly aligned  $\text{TiO}_2$ -rGO hybrid paper electrode after 1000<sup>th</sup> cycle ( $69.3 \Omega$ ) is much smaller than that of the original electrode before cycling ( $410.5 \Omega$ ), indicating the better electrical contact between electrode and electrolyte due to electrolyte penetrating and activation during cycle process.<sup>[40]</sup> The inclined line in the low-frequency region of EIS corresponds to the Warburg impedance ( $W$ ), which is associated with  $\text{Li}^+$  diffusion in the electrode.<sup>[52,53]</sup> Figure 6d shows a good linear relationship between the real impedance ( $Z'$ ) and reciprocal of the square root of the angular frequency ( $\omega^{-1/2}$ ) in the low frequency region. The Warburg impedance coefficient ( $\sigma_w$ ) can be obtained from the slope of  $Z'$  vs.  $\omega^{-1/2}$  in the Warburg region



**Figure 6.** a) Cyclic voltammetry curves of the planarly aligned  $\text{TiO}_2$ -rGO paper at various scanning rates of 0.1, 0.2, 0.4, 0.6, 0.8, 1, 1.5, and  $2 \text{ mVs}^{-1}$ . b) Linear response of the  $I$  as a function of the square root of the scanning rate ( $v$ ). c) Nyquist plots and the fitted equivalent circuit. d) Linear fittings between  $Z'$  and the reciprocal of the square root of the angular frequency in the low-frequency region.

and the square of  $\sigma_w$  is inversely proportional to the  $\text{Li}^+$  diffusion coefficient ( $D_{\text{Li}^+}$ ), that is  $D_{\text{Li}^+} \propto 1/\sigma^2$  (see the Supporting Information).<sup>[52]</sup> The calculated  $D_{\text{Li}^+}$  value of lithiation in the planarly aligned  $\text{TiO}_2$ -rGO hybrid paper is about  $2.89 \times 10^{-13} \text{ cm}^2 \text{ s}^{-1}$  which is one order of magnitude larger than that of the physically mixed paper of  $2.68 \times 10^{-14} \text{ cm}^2 \text{ s}^{-1}$ . The larger  $D_{\text{Li}^+}$  and smaller  $R_{ct}$  of the planarly aligned  $\text{TiO}_2$ -rGO hybrid paper electrode implied faster  $\text{Li}^+$  and electron transfer, consequently resulting in higher power performance, larger capacity, and better cycling stability. The larger  $\text{Li}^+$  diffusion coefficient in the planarly aligned  $\text{TiO}_2$ -rGO hybrid paper electrode is attributed to the special design in which the ultrathin  $\text{TiO}_2$  nanosheets with large exposed (001) facets are face-to-face hetero-assembled onto the rGO current collector to ensure  $\text{Li}^+$  diffusion along the [001] direction of the  $\text{TiO}_2$  thereby yielding excellent rate capability. This layer-by-layer electrode design can be extended to prepare other analogous types of electrodes for high-power LIB and SIBs.<sup>[54]</sup>

### 3. Conclusions

In summary, [001] preferentially-oriented 2D  $\text{TiO}_2$ -rGO nanosheets planarly aligned paper electrodes are designed and fabricated as binder-free, large-capacity, high-rate, and long cycling-life anodes in LIBs. In such spectacular configuration of this planarly aligned electrode, the ultrathin and (001) facets exposed  $\text{TiO}_2$  nanosheets are face-to-faced hetero-assembled onto rGO and the  $\text{Li}^+$  diffuse along the electric field direction can fully intercalate into the (001) facets of anatase  $\text{TiO}_2$  with a low barrier and short transfer distance. At the same time, electrons move quickly along the rGO nanosheets. The planarly aligned  $\text{TiO}_2$ -rGO hybrid paper electrode shows a high specific capacity of  $132.3 \text{ mA h g}^{-1}$  with 89% capacity retention over

1,000 cycles at a large current density of 20 C. It exhibits excellent rate capability with 63% retention when the current density is increased 40 times from 0.5 to 20 C. This layer-by-layer electrode design by coupling different 2D materials with strong synergic effect can be extended to prepare other analogous types of electrode materials for high-power LIBs.

## Experimental Section

### Synthesis of TiO<sub>2</sub> Nanosheets

TiO<sub>2</sub> nanosheets with exposed (001) facets were prepared by a modified hydrothermal route.<sup>[28,29]</sup> In a typical synthesis, tetrabutyl titanate [Ti(OC<sub>4</sub>H<sub>9</sub>)<sub>4</sub>, 12 mL] and hydrofluoric acid solution (3 mL, 50 wt.%) were mixed and then transferred in a 40 mL Teflon-lined autoclave for reaction of 24 h at 200 °C. The as-synthesized TiO<sub>2</sub> nanosheets were rinsed with 0.5 M NaOH solution and deionized water for several times and finally dried overnight in a vacuum oven at 80 °C.

### Synthesis of Freestanding TiO<sub>2</sub>-rGO Planar Paper Electrodes

The GO nanosheets are purchased from Sixth Elements Hi-tech Development Co. Ltd, Changzhou, China. The freestanding TiO<sub>2</sub>-rGO planarly aligned paper electrodes were prepared similar to our previous reports.<sup>[30–32]</sup> In a typical synthesis, the as-prepared TiO<sub>2</sub> nanosheets were dispersed into the poly (diallyldimethylammonium chloride) (PDDA) solution under stirring to be positively charged by the absorption of PDDA cations. The GO solution with a concentration of 2 mg mL<sup>-1</sup> was prepared by probe sonicating as-received GO powders. For the preparation of freestanding TiO<sub>2</sub>-rGO planarly aligned hybrid electrodes, the positively charged TiO<sub>2</sub>-PDDA nanosheets suspension was dropwise added into negatively-charged GO solution dispersion under ultrasonication and then vacuum-filtered through a filter membrane (220 nm pore size). The TiO<sub>2</sub> and GO nanosheets were prone to layer-by-layer self-assembly onto the filter membrane to afford a hybrid film. After freeze drying overnight, the freestanding film of planarly aligned TiO<sub>2</sub> and GO nanosheets was peeled off from the filtration membrane, followed by heating under Ar in a tube furnace at 400 °C for 3 h. During thermal treatment, the GO is reduced to form rGO nanosheets via removing hydroxy groups. Finally, freestanding superlattice-like TiO<sub>2</sub>-rGO planarly aligned paper electrode was produced. For comparison, we also prepared the randomly arranged TiO<sub>2</sub>-rGO electrode by physical mixing TiO<sub>2</sub> and GO nanosheets, vacuum filtration and following thermal treatment in Ar at 400 °C for 3 h. The bare TiO<sub>2</sub> electrode was prepared by mixing the TiO<sub>2</sub> nanosheets, carbon black and poly(vinyl difluoride) (PVDF) at a weight ratio of 80:10:10 and pasted on pure Cu foil.

### Materials Characterization

The morphology, microstructure and composition of the as-synthesized products were characterized by field emission scanning electron microscopy (FE-SEM, FEI Nova 450 Nano), transmission electron microscopy (HR-TEM, FEI Titan G2 60-300) attached with X-ray energy dispersion spectrum (EDS), X-ray diffraction measurement (Philips X' Pert Pro, Cu K $\alpha$  radiation,  $\lambda = 1.5418 \text{ \AA}$ ) and Raman spectrum (HR RASLab). The thermal analysis was performed on a thermogravimetric analysis (TGA, NETZSCH; TG 209 F3) under Ar/O<sub>2</sub> (10% O<sub>2</sub>) atmosphere with the heating rate of 10 °C/min.

## Electrochemical Measurements

The electrochemical properties of the products were studied using 2025 coin cells. The electrolyte was a 1 M solution of LiPF<sub>6</sub> in a 1:1 vol/vol mixture of ethylene carbonate (EC) and diethyl carbonate (DEC). A Celgard 2400 film was used as the separator and pure Li foil as the counter electrode. The galvanostatic charging–discharging (GCD) measurement of the batteries were conducted in a potential range of 1–3 V vs. Li/Li<sup>+</sup>. The specific capacity was calculated based on the mass loading of TiO<sub>2</sub> nanosheets in the electrode. Cyclic voltammetry (CV) was conducted on the CHI 660e electrochemical workstation and the electrochemical impedance spectra (EIS) were acquired before and after cycling 1000 at fully charged state at the rate 20 C via a Princeton applied research (Parstat4000) between 10 mHz and 10<sup>5</sup> Hz.

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## Conflict of Interest

The authors declare no conflict of interest.

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